A REVIEW OF LOW ENERGY SPUTTERING THEORY AND **EXPERIMENTS**

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Abstract

Sputtering mechanisms relevant to the erosion processes of electron bombardment xenon ion engines are described. A review of practical semi-empirical formulations applicable for slow, heavy ions is proposed, along with the existing data for the sputtering o f molybdenum by very low-energy xenon ions. No experimental data are available under 100 eV. Finally, seven different types of experimental techniques for low-energy sputtering measurements are reviewed

1

1. Introduction

On-going efforts to assess xenon ion engine service life are hindered by the lack of qualitative and quantitative knowledge of the physical phenomena governing ion engine failure modes. Brophy et al.' identified seven distinct failure modes for the NSTAR (NASA Solar Electric Propulsion Technology Application Readiness) engine.

Two of these identified failure modes involve direct impact of unfocused primary xenon ions on the molybdenum screen grid webbing. One mode occurs with structural failure of the screen grid, while the other is caused by structural failure of the accelerator grid duc to direct ion impingement from beamlets defocused by flakes of sputtered material on tbc screen grid, which itself is believed to be the primary source of sputtered material.' Another of the identified failure modes results from cathode orifice plate erosion, which involves sputtering by ions with kinetic energies having a DC component between 12 and 20 CV for NSTAR.

kinetic energy of the incident ions corresponds to the engine discharge voltage. Despite successive reductions of discharge voltage values over the past years to the present value of 25 V for NSTAR in an effort to reduce the sputter erosion rates, the impingement of these low-energy ions is still believed to potentially lead to structural failure of the screen grid or the production of sputter deposited films which in turn arc responsible for the formation of metallic flakes that migrate in the chamber and can interfere with the optics. This erosion process can compete with the higher-energy sputtering erosion of the accelerator grid by charge-exchange ions because of a much higher incident ion current density.

In the two cases involving screen grid erosion, the

However, the extremely long duration of life demonstration tests (8000- 12000 hr.) and their elevated cost make it impractical to rely on testing alone to predict ion engine service life. In addition, even the level of confidence one has in the interpretation of tests involving different engine designs or different operating modes is dependent upon the understanding of the physical processes involved.' In particular, knowledge of the low-energy sputtering yield for the incident ions on the screen grid material is necessary in order to analyze and predict the occurrence of screen grid failure or flake formation.

A great amount of research has been motivated over the last five decades by work on subjects such as surface cleaning, sputter deposition of thin films or etching by

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sputtering, and has been carried out on high-energy (greater than a few keV) sputtering yields. In particular, the sputtering mechanism for light ions received attention from attempts at understanding the erosion of planetary surfaces by the solar wind^{2,3} or investigations of the magnetic confinement of hydrogen plasmas for fusion reactors. However, little is known about very low-energy sputtering by heavy ions. This lack of data results in the use of extrapolations from high-energy measurements or semi-empirical formulae to generate estimates for the wear-out rate that are inherently uncertain.

In this paper we present an overview of the physical descriptions of low-energy sputtering and measurement results or attempts. We also describe the different methods available for generating approximate values.

2. Low Energy Sputtering: A Description

The interaction between an incident particle and a solid target can give rise to many different phenomena: the particle can be backscattered, stay in thermal equilibrium with the surface before being subsequently evaporated, excite electronic transitions and provoke the ejection of gamma (secondary) electrons or the modification of chemical bonds, dislodge atoms from the solid surface or even, at very high particle energies, cause radiation damage. The kinetic energy of the incoming particle determines, for the most part, which phenomenon actually occurs or dominates.

Physical sputtering is an atomic scale **process'** that can occur if the incident particle (ion) can transfer sufficient energy to a surface or bulk target atom to overcome its bulk displacement energy anti/or its surface binding energy. The erosion due to physical sputtering is described by the sputtering yield, Y, a statistical variable defined as the mean number of atoms removed from a solid target per incident ion. Sputtering by elastic collisions can have three regimes: the single-knockon, the linear cascade or the spike regime.⁶

The processes involving a linear collision cascade or the spike regime become less important at energies near threshold. Behrisch⁷ summarized the possible sputtering mechanisms for low-energy light ions as reproduced in Fig. 1. These mechanisms arc in fact still possible for heavy ions on light targets, as is most often the case with xenon, with the provision that the processes involving an outgoing ion (S_{II}) arc less probable than with light ions since backscattering of a heavy ion (impossible for a headon collision) will demand more collisions. In the case of normal ion incidence, a minimum of two collisions are necessary for producing a sputtered atom. In turn, atoms sputtered as a result of very few low-energy collisions are likely more be sputtered at incidences.8,9

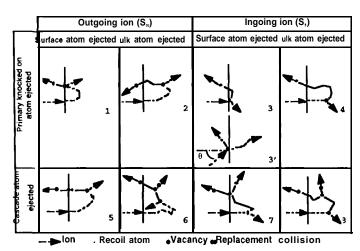


Fig. 1: A "billiard-ball" description of possible sputtering mechanisms

A pure hard-sphere, classical two-particle elastic scattering model is often used to describe physical sputtering, but the energy transfer between ions and atoms or between two cascade atoms is complicated by the complex electronic screening of the two nuclei," In the limit of high-energy particles, the collision kinematics arc the same as in Rutherford scattering of two point charges, but for low-energy collisions a detailed description would require taking into account the physics of quantized screened Coulomb collisions and the absorption of energy by Pauli promotion of the electrons.") Finally, the nuclear stopping power, i.e., the probable energy loss of the ion per unit distance traveled through the target, depends on which classical atomic model is chosen.") Unfortunately, these models give significantly different results at low energy, as shown on the plot reproduced for convenience in Fig. 2.

UNIVERSAL Reduced Nuclear Stopping

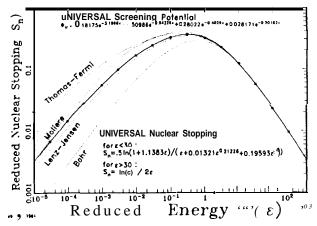


Fig. 2: Nuclear stopping power using the universal screening potential (solid line) or the four classical atomic models (from Ref. 10).

It should also be noted that at energies near threshold a new difficulty arises, due to the fact that the Rutherford scattering cross section increases with decreasing particle velocity. This in turn implies that the collisions can no longer be treated as independent binary collisions, but rather involve several neighboring atoms. ^{5,11,12}

Threshold Energy And Atomic Mass Dependence

Since physical sputtering is a collision process, it is intuitive that it should be a function of the atomic mass of the collision partners. It can easily be shown (see e.g. Ref. 10) that the fraction of energy transferred to a target particle of mass m_2 by a projectile of mass m_1 in a head-on (zero scattering angle) collision is, if $m_1 > m_2$:

$$\gamma = \frac{E_2}{E_0} = \frac{4m_1 m_2}{(m_1 + m_2)^2} \tag{1}$$

where γ is the energy transfer factor, E_2 is the kinetic energy transferred to the target particle and E_0 is the initial kinetic energy of the projectile.

Bradley' concluded from this and early experiments in 1954 that the threshold energy for sputtering could be predicted by:

$$\frac{U_S}{E_{th}} - \frac{4m_1 m_2}{\left(m_1 + m_2\right)^2} \tag{2}$$

where U_s is the atomic heat of sublimation and E_{th} the threshold energy. This was later confirmed by Wehner¹² who, in 1958, suggested that in a first approximation E_{th} was proportional to U_s/γ , with the proportionality factor being between 8 and 20, and pointed out that sputtering presented quite the same periodicity with atomic number as the heat of sublimation.

A somewhat counterintuitive result was however published in 1962 by Stuart and Wehner, ¹⁴ who first realized that in fact the mass ratio between ion and target atom played hardly any role in the thresholds. This remark was then confirmed by Wehner and Anderson^y who suggested as an explanation that at low energy the collisions could not be treated as independent successive two-particle collisions and recommended that "modified masses" be introduced. The fractional energy transfer γ was therefore dropped from some models used to predict the sputtering threshold, but E_{th} was still subsequently approximated as $4U_S$ for heavy ions⁴ or both for heavy and light ions, ${}^5U_S/\gamma$ for light ions 15,16 or for $m_1 < m_2$, ${}^{17}U_S/\gamma(1-y)$ for $m_1/m_2 \le 0.3$, ${}^{17}8U_S(m_1/m_2)^{2/5}$ for $m_1/m_2 > 0.3$, 17

$$E_{th} = 1.5 \frac{U_s}{Y} \left[1 + 1.38 \left(\frac{m_1}{m_2} \right)^h \right]^2$$
 (3)

where h=0.834 for $m_2 > m_1$ and h=0.18 for $m_2 < m_1$, ¹⁸ or finally:

$$E_{th} = U_s \left[1.9 + 3.8 \left(\frac{m_2}{m_1} \right)^{-1} + 0.134 \left(\frac{m_2}{m_1} \right)^{1.24} \right]$$
 (4)

where Eq. 4 is from Ref. 19. For completeness, it should be added that Weissman and Sigmund" suggested $E_{th} \approx U_s$ and Olson et al. ypointed out that the very poorly defined sputtering threshold (identified to the surface binding energy) may have an effective value much less than the heat of sublimation. Which approximation is to be used, except for Eq. 4, depends on which sputtering mechanism dominates (see Fig. 1): as pointed out by Weissmann and Behrisch, 20, sputtering by light ions is primarily driven by backscattering of the ions from the interior of the target (reflective collision process), whereas for heavy ions col! ision cascades generated by direct impingement of the incoming ions dominate the sputtering mechanism (displacement process). This distinction is extremely important for the threshold energy, as well as for the angular dependence of the low-energy sputtering yield.23

Incidence Angle Dependence

Ref. 21 provides an excellent background as well as an extensive list of references on the subject of angular dependence of sputtering yields that will not be duplicated here. Yamamura $et\ al.^{21}$ reported that numerous investigations showed the angular dependence of the sputtering yield to behave like $cos^{-1}\theta$ for not-too-oblique incidence, where θ is the angle of incidence measured from the surface normal, while Sigmund²² obtained from theoretical studies a dependence in $cos^{-1}\theta$, where 1 < c < 1 Yamamura $ct\ al.$ 19 pointed out that the threshold energy for heavy-ion sputtering was mainly determined by the anisotropic velocity distribution of the recoil atoms, and Yamamura $ct\ al.$ 19 further indicated that this was also the reason why the threshold energy had a minimum near 60, unlike in the case of light-ion sputtering.

An empirical formula for the angular dependence of sputtering can be given as:²¹

$$\frac{\mathbf{Y}(\mathbf{0})}{\mathbf{Y}(\mathbf{0})} = \cos' \mathbf{f} \, \boldsymbol{\theta} \exp_{l} - \Sigma \left(\cos^{-1} \boldsymbol{\theta} - 1\right)] \tag{5}$$

where f and Σ are energy-dependent fit parameters as given in Ref. 21, with values 19.96 and 13.55 respectively for 100-CV xenon ions on molybdenum. Y(0) is the sputtering

yield at normal incidence. The exponent f carries the threshold effect and is a function of the ratio Z-YE,},. The angular dependence of E_{th} itself is driven by the superposition of the threshold energy for the sputtering process S, in Fig. 1 and the threshold energy corresponding to surface channeling,²¹ i.e.

$$E_{th}(\theta) = E_{th}(0)\cos^{2}\theta + \frac{0.3 - \frac{A}{A+1}E_{Th}\left(\frac{a}{R_{0}}\right)^{3}}{\cos^{2}\theta}$$
 (6)

where $E_{th}(0)$ is the threshold energy for normally-incident ions, A is the mass ratio m_2/m_1 , the Thomas-Fermi screening radius a is a function of the atomic numbers Z_1 and Z_2 of the projectile and the target respectively and is given by

$$a = 0.4685 \left(Z_1^{2/3} + Z_2^{2/3} \right)^{-1/2}, \tag{7}$$

 R_0 is the average lattice constant of the target, given by $R_0 = N^{-\frac{1}{3}}$ where N is the number density of the target atom, and the Thomas-Fermi energy unit E_{TF} is given by

$$E_{TF} = \frac{m_1 + m_2}{m_2} \frac{Z_1 Z_2 e^2}{a} \tag{8}$$

from 1.SS theory (see for example Refs. 6, 10 or 23). e is the proton charge. The first term on the right-hand side of Eq. (6) corresponds to the sputtering mechanism SI (see Fig. 1), while the second term describes the onset of surface channeling and dominates for large values of θ . Fig. 3 shows the threshold energy as a function of incidence

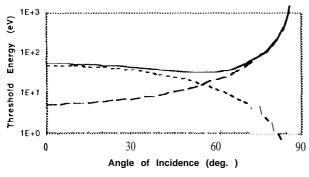


Fig. 3: Angular dependence of the threshold energy. The dotted line is the threshold energy for the sputtering mechanism S_I , dominant for heavy ions at not-too-oblique angles, while the dashed line is the threshold energy corresponding to surface channeling, valid for grazing angles o f incidence. The solid line is the sum of the two contributions'.

angle for these two mechanisms for xenon ions on molybdenum. The net sputtering threshold energy exhibits a minimum near 60.

3. Predictions for Low-Energy Sputtering

Empirical Formulae

In order to predict the ion erosion rate on the NSTAR engine, Rawlin²⁴ used a quadratic approximation to extrapolate the energy dependence of the sputtering yield in the region near threshold. Other authors have developed or used empirical or semi-empirical models.

Sigmund²² proposed a systematic study of the sputtering of a random monoatomic target in the linear collision cascade regime based on transport theory. The Sigmund equation,

$$Y(E) = \frac{0.042 \alpha (M_2/M,) S_n(E)}{U_c}$$
(9)

gives the energy dependence of the yield Y(E) as a function of the measured sublimation energy U_s , the elastic (nuclear) stopping cross section Sri(E), and the fit parameter $\alpha(M_2/M_1)$. Matsunami et $al.^{25}$ adapted this formula by taking into account the effect of the threshold energy to write the first Matsunami formula:

$$Y(E) \quad \frac{0.42 \, \alpha K s_n(\varepsilon)}{U_s} \left[1 - \left(\frac{E_{th}}{E} \right)^{V_2} \right] \tag{10}$$

where $\varepsilon = E/E_{TF}$ is the reduced energy, $s_n(\varepsilon)$ is Lindhard's elastic reduced stopping cross section and K is the conversion factor from $s_n(\varepsilon)$ to $S_n(E)$, as defined in Ref. 25. Yamamura et al.²⁶ further refined this equation by making the inelastic stopping explicit in:

$$Y(E) = \frac{0.042 \alpha S_n(E)}{U_s [1 + 0.35 \ U_s - S_r(E)] (E)} \frac{1}{(E)} \frac{|E_{th}|^{1/2}}{|E|}^2$$
(11)

where $S_{\epsilon}(\mathcal{E})$ is Lindhard's reduced inelastic (electronic) stopping cross section. Finally, Yamamura et a/.27 rewrote Eq. (11) to take into account the effect of the target material on the mass-ratio dependence, i.e. by substituting $\alpha(M_1/M_1)$ by $Q(Z_2)\alpha^*(M_1/M_1)$ to yield the third Matsunami formula:

$$Y(E) = \frac{0.042 \quad Q\alpha^* S_n(E)}{U_s[1 + 0.35 \quad U_s S_e(E)]} \left\{ \frac{E_{th}}{E} \right\}^{\frac{1}{2}} \left\{ (12)$$

Appropriate tables and definitions to obtain the numerical values of the different parameters for 259 ion-target combinations are given in Ref. 28.

Similarly, Bohdansky developped both a formulation valid only for light ions¹⁷ and the universal relation below:²⁹

$$Y(E) = \frac{0.042 \,\alpha S_n(E)}{U_s} \left(\frac{R_p}{R}\right) \left[1 - \left(\frac{E_{th}}{E}\right)^{\frac{2}{3}}\right] \left[1 - \left(\frac{E_{th}}{E}\right)\right]^2 \quad (13)$$

where E_{th} takes the value $\delta U_s(m_t/m_2)^{2/5}$ and R_{tt}/R is given by:

$$\frac{R}{R_P} = 0.4 \ M_2 / M_1 + 1 \tag{14}$$

if inelastic losses are neglected.24

A plot showing the energy dependence of the sputtering yield of molybdenum with xenon ions using the formulation given by the third Matsunami formula and Eq. (13) is shown in Fig. 4. Results of computer calculations with the Monte-Carlo simulation code TRIM and experimental data are also shown.

The third Matsunami formula or Eq. (13), along with Eq. (5), are helpful analytical expressions, and were used by Bond and Latham³⁰ in the plasma simulation code SAPPHIRE to calculate grid erosion rates due to charge-exchange ions in the UK-10 ion thruster. Sigmund²² however noted in his theory that the surface collisions that

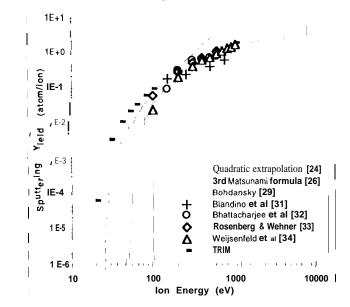


Fig. 4: Energy dependence of the sputtering yield of molybdenum with xenon ions.

dominate sputtering near threshold cannot be described by transport theory, and that the concept of binary collisions that he assumed becomes questionable at low energy. Extreme care should therefore be used with Eqs. (9)-(12) at energies near threshold.

4. A Review' of Experimental Methods

Driving Requirements

Measuring sputtering yields for materials of interest in ion thruster technology with slow incident ions raises very serious experimental difficulties, 14,35 listed below:

The vanishingly small amounts of sputtered material to be measured in a practical exposure time mandates the use of an extremely sensitive method and requires as high an ion current density as possible.

- The ion beam should have as low an energy spread and as low a divergence angle as possible, to insure that the ion energy and incidence angle are well known and controlled, This implies in particular that the multiply charged ion current be as small as possible, which requires a low discharge voltage. Another concern regarding the beam could be the presence of fast neutrals due to charge exchange collisions in the ion source. The choice for the method of generating the ion beam usually results from a trade off between the relatively high current densities achievable with a plasma discharge (typically up to 15 mA/cm²) and the better-defined beam that can be obtained with an ion gun, in terms of energy, incidence angle and impurities. Ion guns however arc limited in beam current density due to the space charge limitation, proportional to $V^{3/2}$ where V is the accelerating voltage.

A low background gas pressure needs to be achieved in the facility in order to mitigate the formation of a protective chemisorbed impurity layer at the target sur-face. This problem is also related to the ion beam current density and is described in more detail below. A low base pressure is also required insure that the mean free path of the ions is larger than the source-to-target distance, so that uniformity of the beam in energy and incidence angle is preserved,

Effects of Background Gases on Sputter Erosion in Ion Thrusters

Lifetests of ion thrusters to measure the screen grid sputter-erosion may be seriously compromised by the effects of background gases. The sputtering rate of the screen grid has been found to be reduced from that of a dynamically clean surface by factors up to eight. If vacuum conditions are not adequate in the test facility, the various species of background gases will be chemisorbed on the screen grid

surface and will act as a buffer to the impinging ions, thus reducing the net erosion rate of the screen grid. This effect is shown in Fig. 5. The plasma in the discharge chamber excites the background gas molecules, which dissociate upon colliding with the target surface increasing the reactivity of the background gases. When the pressure is sufficiently low, the sputter-erosion rate is equal to that of a dynamically clean surface. As the background pressure is increased, chemisorption on the screen grid surface will begin to take place. If the background pressure reaches still higher levels, the sputter-erosion will level off as compound or compounds of the target and the background gases³⁸ arc formed. Under these conditions, the sputtering of the compound or compounds wiii take place instead of a clean surface. 'I'he pressure at which chemisorption wiii begin to affect the measured sputtering rate depends on the thruster operating condition, the discharge voltage which determines the ion energy, and the current density of the singly and doubly charged ions.

To determine the proper vacuum conditions at which the effect of the background gases becomes negligible for the operating conditions of the thruster is no easy task. It has often been stated that a sufficient condition for a dynamically clean surface is that the flux of the impinging ions be larger than the flux of the impinging gas to the target area. This, however, is not a sufficient condition. The sputtering rate of the absorbed species has to be taken into account. The condition to ensure a dynamically clean surface is:

$$\frac{Y_{i,i}I}{f\beta_{i}} \ge 10 \tag{15}$$

whet-c:

 $Y_{i,s}$ = sputtering yield of species i absorbed on target s

I = ion flux impinging on target sf = background gas flux impinging on target sb_{i,s}= sticking probability of background gas species i on a clean target s

The parameters most difficult to determine arc $Y_{i,s}$ and $b_{i,s}$. A theory for the sputtering of chemisorbed gases has been developed and the formulations applied to thruster lifetime tests met with some degree of success. It has been demonstrated that most background gases found in diffusion pump vacuum facility such as N_2 , H_2O , O_2 and C_2N_2 , will react with the target and be chemisorbed on the target.

To calculate the ratio defined by Eq. (15), it is assumed that the bulk of the chemisorbed gas on the screen grid target is nitrogen. In a common diffusion pump vacuum facility in which an air leaker liquid nitrogen trap leak is the most probable source of background gas, nitrogen is the largest constituent of the background gases. The sputtering yields of chemisorbed nitrogen were calculated using Winter's formulation. 41 For NSTAR conditions, the sputter yield of nitrogen for singly charged xenon ions at 25 V is 0.02 atom/ion and for doubly charged xenon ions, the yield is 0.08 atom/ion. The sticking coefficient was assumed to be equal to 1.0 because of the high reactivity of the species due to the presence of the discharge plasma. The ratio defined by Eq. (15) is estimated to be 5 for the given test conditions, which is somewhat lower than the recommended value of 10. Therefore, some reduction of the sputtering rate of the screen grid is to be expected in a lifetest due to background gases, if the operating pressure of the vacuum facility is -2.0 x 10-' Torr. The magnitude of the reduction is not well defined.

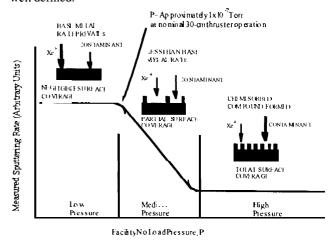


Fig. 5: Sputtering rate as a function of facility no load pressure

Measurement Techniques and Results

A brief review of experimental techniques follows, and is summarized in Table 1. Seven families of methods for measurement of sputtering yield have been identified, Other early methods from the 1920's through 1950's, some of which were reasonably sensitive, are not listed here but arc described in Ref. 14.

The first and most extensively used method involves measuring the weight loss of the eroded sample. 12,16,33-35,43-46 While this method allows fOr absolute, direct measurements on all materials and is relatively simple, its sensitivity, limited to about 10⁻³ atom/ion makes it inappropriate for sputtering at energies near threshold.

The same remarks apply to profilometry.^{31,47,48} This method involves measuring the depth of the sputtered surface with a micro-stylus, with reference to a masked area

Method	Comments	Ref.	Ion/Target Combination	Ion Beam Generation	Ion Energy Range (eV)	Beam Current Density (mA/cm²)	Background Pressure (Torr)	Lowest Sputtering Yield Measured (atom/ion)
Weight Loss	-Direct, versatile, absolute values	33	Hg ⁺ on 26 metals	Plasma disch.	30-400	-3 (up to 15)	-10	-1 X1 O'''
		12	Hg ⁺ on metals and semiconductors	Plasma disch.	50-400	-5 (up to 15)	~ 10.6	-0.2
		41	Ar ⁺ and Ne ⁺ on metals and semiconductors	Plasma disch	50-600	2-15	-7 XI O'''	4X IO"'
		31	Ni ⁺ , $\overline{A}r^+$, Kr^+ and Xe^+ on Cu, Ni, Fc and Mo	Plasma disch.	1 oc- 1000	-1.5		-0.1
		30	He+, Kr+, and Xe+ on various metals	Plasma disch.	100-600	2-15	10'''- 10'''	1x10-5
	-Low sensitively	42	H ₂ ⁺ on 304 SS, pyre. graphite, C fibres, glassy C and SiC	Ion source	500- 7,500	Up105	10.7	3X IO"
		16	H+, D ⁺ , He ⁺ and He ⁺ on Mo and Au	Ion Source	150- 20,000	0.5 to 1	~10-7	IX10 ⁴
		43	Selfsputtering for Ni	Plasma disch.	75-3,000	0.		7x10 ⁻²
		44	H ⁺ and He ⁺ on Ni and W	Ion source	1,000- 4,000	0.5 to 5	-10'	2. lx 10 ·
Profilo-	-Direct, versatile, absolute values	47	Ar' on Si, Ag, Cu, Ni, Ti and Al	lon gun	10,00Q	0.1		~10 ⁻² (inferred)
metry	-Low sensitivity	31, 48	Xc' on CVD diamond, C-C and Mo	Ion source	1s0-750	0.3-2.6	~10.9	0.1
Optical, Trans- mission	-Versatile, <i>In situ</i> -Relative values, indirect	49	Hg⁺ on metals	Compressed plasma disch	20-200	up to 500		3x10 ⁻⁴
Surface Ionization	-Sensitive, direct -Limited to alkali metals, relative values	13	He ⁺ , Ne ⁺ , Ar ⁺ and Xe ⁺ on Na and K	Ion gun	0-1,800		5x10 ⁻⁸	3x10 ⁻³
Radio- active Tracer	-Very sensitive, direct -Few suitable isotopes,	50	Hg+ and Ar+ on Co	Plasma disch	10-100	-	-	1x10 ⁻⁴
		51	Ar+ and Xe+ on Co, Cd and Cr	Ion gun	10-s00	up to 0.1	2X1 O'''	2X1 O'''
	relative values,	52	Ar and Xe on Cr	Ion gun	50-500	0.01-0.1	2x10 ⁻⁷	5x10 ⁻²
	complex	53	Ar ⁺ and Xe ⁺ on Co and Cr	Ion gun	10-50	0.002-0.02	2X IO"	4x10 ⁻⁵
QCM	-Very sensitive, absolute, simple, direct, in situ	55	Ar ⁺ on Au	I on source	0-100	-0.01	1x10-7	3x10 ⁻³
	-Requires sample material to be coated as thin film on crystal	56	Sefsputtering for Au, Cu, Ag, Cr and Al	Ion gun	10-500	-	2X10"7	-10' (1.1 in less than 10 sec.)
		57	Hg ⁺ and Ar ⁺ on Cr	Plasma disch.	25-300		474.00	10"
Spectro- scopic	-Very sensitive, versatile, in situ	14	Nc', Ar ⁺ , Kr', Xc ⁺ and Hg ⁺ on 23 metals	Compressed plasma disch.		40 (up to 100)	1X1 O'''	10'
Methods	-Relative values, indirect	59	Ar ⁺ on Ag and Nb	Ion source	500- 1,500	0.08 (estimate)	1.5 X1 0" w/o bakeout	0,2 measured 10" possible
		32	Xc' on Mo	Ion gun	150-600	0.03	2X10'	0.1

Table 1: A Review of Experimental Techniques for pleasuring Low Sputtering Yields

of the target. One difficulty with this method is that its precision depends on the eroded depth profile roughness.

Askerov and Sena⁴⁹ used the change in the optical transmission of the plasma radiation to a photoresistor through the sputtered film deposited on a glass wall. Although this method has the advantage of being *in situ*, it is indirect, gives a relative value for the sputtering yield and is still of a too low sensitivity for very slow ions.

Another somewhat original and relatively sensitive method was described by Bradley, ¹³ who used the property that alkali metal atoms lose their valence electron when striking a hot metal surface. In these experiments, a negatively-biased electrode collected the ion current corresponding to the sputtered alkali atoms ionized upon impact on a hot platinum surface. This method is

unfortunately confined to alkali metals and only gives relative sputtering coefficients.

Methods using radioactive tracers were proved by Morgulis and Tishchenko⁵⁰ and Handoo and Ray⁵¹⁻⁵³ to provide a great benefit in sensitivity, enabling measurements in the near threshold region. These methods however presents the inconvenient of requiring a suitable isotope for the material to be sputtered (in terms of half life and energy of the gamma-ray emissions) or the activation of a surface layer of the sample in a specialized facility.⁵⁴

Another very sensitive method, using Quartz Crystal Microbalance (QCM) techniques, was early proposed by McKeown.⁵⁵ With modem-day frequency-measurement technology, using a QCM could theoretically enable to measure directly, *insitu*, sputter yields as low as

10s atom/ion even at low current density. A difficulty associated with this method is that it requires the sample material to be coated as A thin film (a few um thick) cm the quartz crystal.

The last family of measurement techniques encompasses spectroscopic methods such as optical spectroscopy, '487,5x Auger Electron Spectroscopy (AES), 59 Rutherford Backscattering Spectroscopy (RBS), Secondary Ion Mass Spectrometry (SIMS), or Secondary Neutral Mass Spectrometry (SNMS). 32 Optical spectroscopy, RBS, SIMS and SNMS arc all indirect methods and can only give relative sputtering yields. RBS was used by Bhattacharice et al. 32 to calibrate measurements obtained with SNMS or SIMS, more sensitive.

5. Current work and Conclusion

Upon reviewing the literature available on low energy sputtering yields, the authors have concluded that:

- (a) There is currently no satisfactory data available for the sputtering yield of molybdenum by xenon ions under 100 eV to adequately assess ion engine service life by analysis.
- (b) Extrapolations and semi-empirical formulae are available to obtain rough estimates with an uncertainty probably over 50%. They are based on fit parameters obtained with existing, higher-energy data.
- (c) Empirical formulations or computer simulations are based on physical theories that break down at low energy because the interatomic potentials and the assumptions on the nature of the collisions no longer represent an accurate description of the reality.
- (d) The measurement of sputtering yields with energies near threshold poses serious difficulties, essentially due to contamination issues and sensitivy requirements.

modern instruments, However. measurement techniques and vacuum facilities might enable some significant progress in the near future. Experimental work is ongoing at Tuskegee University with spectroscopic methods and JPL with piezoelectric QCM.

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